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Overview of corrosion behaviour of ceramic materials in molten salt environments

ABSTRACT

This study reviewed previous studies between the years 2015 and 2021 on how ceramic materials degraded in the presence of molten salt environments. The processes of corrosion resistance of various ceramic compositions subjected to various molten salt compositions and temperatures were also scrutinized. The results offer important new insights into the variables affecting ceramics' corrosion behaviour and the production of corrosion products. The reported result reveals that the ceramic material with the composition (Smo.5Sco.5)2Zr2O7 performed better than that of Sm₂Zr₂O₇ in terms of hot corrosion resistance in molten salt (V₂O₅ + Na₂SO₄). It has also been reported that corrosion behaviour is influenced by particle size. Notably, zirconia (n-YSZ) with nanoscale grain sizes was more susceptible to hot corrosion, which was explained by increased specific surface areas. On the other hand, sintering and additives have been found to enhance corrosion resistance. The Y-Y₂Si₂O₇ ceramic's resistance to corrosion in (V₂O₅ + Na₂SO₄) molten salt was enhanced by the addition of alumina. The results of these investigations help us understand how corrosion works and what influences ceramic materials' susceptibility to deterioration in molten salt environment. This information can direct the development of more corrosive-resistant ceramic materials for use in high-temperature environments or molten saltbased energy systems, among other corrosive uses.

Keywords: hot corrosion, alumina, heat exchangers, nitrides, zirconia

1. INTRODUCTION

A process called "hot corrosion" occurs when molten salts attack a material's surface at the same time that they interact with the oxide layer that has formed there, often at a high temperature [1]. Hot corrosion is typically considered an accelerated deteriorating process that involves the deposition of corrosive species (like sulphates) from the environment (like combustion gas) to the surface of hot components, followed by the subsequent disintegration of the protective oxide scale [2]. Consequently, corrosive species are created that can enter the material and cause a variety of degradation processes, such as oxide spalling, grain boundary assault, and material embrittlement. Hot corrosion may dramatically lower a material's mechanical strength and lifespan, impairing the functionality and dependability of crucial parts [3].

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In general, the menace of corrosion refers to the degradation of materials occasioned by electrochemical or chemical reactions. Unlike ceramics, metals tend to degrade in acidic environments and previous studies have shown that the corrosion of metals can be deterred in various acidic media using different plant extracts and pastes [4–13]. When exposed to acidic conditions, ceramics display improved corrosion resistance in comparison with metals [14]. In places where ceramics are used, such as heat engines and heat exchangers, molten salt deposits are significant corrodents and it has been reported that the Na₂SO₄-induced corrosion can be quite serious [15].

Ceramics, including structural ceramics, are renowned for having extraordinary qualities including resistance to chemical reactivity, wear, and high temperature [16]. In the commercial world, these ceramics are utilized in a wide range of products, including gas turbine components, welding nozzles, heat exchangers, heat pipes, valves, cutting tools, turbochargers, crucibles, dental work, bone, and joint replacement prostheses [17]. In addition, these promising

qualities make ceramics desirable for a wide range of applications, including heat barrier coatings, refractories, and catalytic supports [18]. However, the hot corrosive behaviour of ceramics in hostile conditions, such as molten salts, might restrict their performance and durability [19]. Designing and creating ceramic materials with better resistance to corrosive conditions, therefore, requires an understanding of the ability to characterize the hot corrosion behaviour of ceramics [20].

The extremely corrosive conditions of molten salts may seriously degrade materials [21]. They are made up of a combination of cations and anions, which can react with ceramic surfaces to cause corrosion [22]. Temperature, composition, and the presence of impurities are all factors that affect the corrosive attack in molten salts [23]. Various corrosive species, including chlorides, sulphates, and fluorides, can be introduced into the molten salt environment and speed up the deterioration of ceramics [24]. Further enhancing the corrosive assault may be the presence of impurities or strong gases.

Over a few decades, studies were done on hot corrosion of Na_2SO_4 -coated Ti_3AlC_2 in the air at $700-1000^{\circ}C$ [25]. No protective scales were reportedly seen, which led to significant hot corrosion of Ti_3AlC_2 by fused Na_2SO_4 at 900 and $1000~^{\circ}C$. The hot corrosion attack was minimal below the melting point of Na_2SO_4 , whereas Ti_3AlC_2 experienced a strong hot corrosion attack from fused sodium sulphate above the melting point of Na_2SO_4 . Another study examined the hot corrosion of Na_2SO_4 -coated and Ti_3SiC_2 -based ceramics at 900 and 1000 $^{\circ}C$ in air [26]. According to reports, the Ti_3SiC_2 ceramics covered with roughly 2 mg/cm² Na_2SO_4 experienced substantial damage.

Research has been conducted on the hot corrosion process of composite Alumina/Yttriastabilized zirconia coating in molten sulphatevanadate salt [27]. Results demonstrate that when the salt's V₂O₅ level was less than around 5 wt%, the Al₂O₃ overlay served as a barrier to prevent the molten salt from penetrating the YSZ coating. Due to this, ZrO₂ instability was greatly mitigated [27]. The hot corrosion behaviour of Ti₃SiC₂ in the combination of Na₂SO₄-NaCl melts has been investigated [28]. When the concentration of Na₂SO₄ was more than 35 wt% at 850 °C, they reported that the Ti₃SiC₂ had a significant hot corrosion attack. There were some radial and transverse fissures in the outer and inner layers of the corrosion layer, which exhibited a duplex microstructure. Xie et al. [29] investigated the hot of double-ceramic-layer behaviour LaTi₂Al₉O₁₉ and yttria-stabilized zirconia thermal

barrier coatings. They reported that the bond coat in the LaTi₂Al₉O₁₉/yttria-stabilized zirconia thermal coatings significantly contributed protecting the component against hot corrosion. In addition, the ceramic material, LaTi2Al9O19/yttriastabilized zirconia was found to show good chemical stability in molten salts of Na₂SO₄ and NaCl. The disintegration of LaTi₂Al₉O₁₉ did not occur even after 1458 hot corrosion cycles at 1373 K, indicating high chemical stability in molten salt of Na₂SO₄ and NaCl, according to further observation [29]. The reason for this behaviour was attributed to the outstanding high-temperature performance and limited heat conductivity of the ceramic material (LaTi₂Al₉O₁₉).

The main objectives of this current study are to:

- 1. Review previous studies on the behaviour of ceramic materials in molten salt environments from 2015 to 2021, to better understand how they degrade and how well they can resist corrosion.
- 2. Identify and examine the variables, such as ceramic material composition, particle size, sintering methods, and salt composition, that affect the corrosion behaviour of ceramics in molten salt, and to
- 3. Identify knowledge gaps and potential directions for additional study in the hot corrosion behaviour of ceramics.

2. BEHAVIOUR OF CERAMICS IN MOLTEN SALTS

2.1. Ceramics attacked by molten salts – Nitrides and carbides

When different molten salts are present, the typically protective layer of SiO2 that develops on SiC and Si₃N₄ can show accelerated corrosion. According to earlier research [30], an acid salt solution did not cause the protective layer to erode. Active corrosion was caused by the production of SiO gas at low oxygen pressures. It has been demonstrated that a key factor in the behaviour of molten sodium salts is the activity of Na₂O. The possible reactivity with silica increases as this activity rises [30]. Lin et al. [31] studied the 850-1000 °C hot corrosion behaviour of Na₂SO₄-coated Ti₂AlC in air. Below the melting point of Na₂SO₄, a protective Al₂O₃ scale was discovered to slow down further corrosion of the Ti₂AIC substrate. The hot corrosion kinetics were found to be parabolic. Because porous and non-protective scales failed to shield the Ti₂AIC substrate from hot corrosion assaults, severe hot corrosion attacks on Ti₂AlC were reported to occur at 900 and 1000 °C. Notably, it was observed that sulphur separated at scale/substrate interfaces and chemical interactions between Ti₂AlC and the Na₂SO₄ salt dominated the corrosion process. The hot corrosion resistance of Ti₂AlC can be improved by limiting the penetration of Na₂SO₄ to the Ti₂AlC substrate, and this was suggested as a practical and effective pre-oxidation treatment [31].

2.2. Ceramics' Hot Corrosion Mechanisms

Studies have shown that hot corrosion in ceramics involves several intricate processes that accelerate material deterioration, especially in the presence of sulphate-related salts [32]. On the other hand, yttria-stabilized zirconia's LZ₇C₃ coating was discovered to be flawless and degrading very little [33]. However, certain ceramics have been reported to exhibit relatively low corrosion when exposed to molten NaCl-KCl-MgCl₂ [34]. Designing solutions to reduce hot corrosion and increase the longevity of ceramic components in corrosive environments requires an understanding of the mechanisms behind the hot corrosion of ceramics.

2.2.1. Degradation owing to chemical processes

One of the main processes causing hot corrosion in ceramics is degradation owing to chemical processes. The ceramic surface may react with molten salts, producing soluble reaction products. The protective oxide layer on the ceramic surface may be removed by these reaction products, exposing the underlying material to more corrosion. The reactivity of the molten salt, the makeup of the ceramic material, and the system temperature all have an impact on the chemical assault process. A study to understand the hot corrosion deterrence of YSZ, Gd₂Zr₂O₇, and $\mathsf{Gd}_2\mathsf{Zr}_2\mathsf{O}_7 \ \ \textbf{+} \ \ \mathsf{YSZ} \ \ \mathsf{composite} \ \ \mathsf{thermal} \ \ \mathsf{barrier}$ coatings in Na₂SO₄ + V₂O₅ at 1050 °C has been conducted [35]. In heated corrosion testing, the authors observed that a series of linked chemical processes might be classified as the chemical degradation of common YSZ coatings. Additionally, it was discovered that NaVO₃ would occur when V₂O₅ and Na₂SO₄ salt combination were exposed to high heat (1050 °C).

2.2.2. Degradation because of diffusion-controlled processes

Concentration gradients cause the transport of corrosive substances into the ceramic in the diffusion-controlled process of hot corrosion. Corrosive species infiltrate into the ceramic from the surrounding environment, and as they do so, their concentration rises. Temperature, concentration gradients, and material characteristics, such as diffusion coefficients, all affect the diffusion process. Costa-Oliveira and Baxter [36] investigated how salt corrosion affected a hot-pressed silicon nitride in combustion circumstances with various sulphur concentrations. It was reported that

depending on their concentration in the gas phase, pollutants including the salts, NaCl and Na_2SO_4 frequently found in gas turbine atmospheres were expected to accelerate the corrosion of Si_3N_4 due to their absorption into the scale. This was clarified by assuming that corrosion in sodium-containing environments happened via an ionic diffusion process [36].

Corrosive species can react with the ceramic's components once they get within the material; for instance, if sulphur is present, it may interact with the metal ions in the ceramic lattice to generate sulphides. The creation of new phases like oxides, sulphides, or halides because of the interaction between corrosive species and ceramic elements might harm the material's characteristics. The porosity, grain boundaries, and flaws of the ceramic material can all have an impact on how easily corrosive species diffuse inside it. Corrosive species can enter through porous materials, which speeds up their transportation. They can operate as preferred diffusion channels for the ceramic material's grain boundaries, which are the interfaces between different ceramic grains. Cracks and gaps in materials can act as entrance routes for corrosive species, hastening the deterioration process. Electrochemical techniques have been used to assess the corrosion effect of zirconia addition at 1, 2, and 5 atomic percents (at.%) on the hot corrosion behaviour of Mo₃Si intermetallic alloy in 60 weight per cent NaNO3 and 40 weight per cent KNO₃ at 600°C [37]. For the as-cast alloy or when 1 or 2 wt.% Zr was added, and the corrosion process was controlled by charge transfer, but when 5 wt.% Zr was added, and it was controlled by diffusion [37].

2.2.3. Degradation by electrochemistry

Electrochemical reactions may occur at the ceramic surface because of molten salts acting as electrolytes. As a result of these interactions, corrosive species are formed by the transfer of ions between the ceramic material and the corrosive environment. The presence of electrochemically active species in the molten salt, the electrical conductivity of the ceramic and the temperature all have an impact on the electrochemical attack process. Li et al. [38] used a molar ratio of 1:1 to study the hot corrosion behaviour of Yb2Zr2O7 ceramic against V_2O_5 and Na_2SO_4 + V_2O_5 at temperatures of 900-1100 °C for 2 hours in the air. They noticed that the YbVO₄, which resulted from the hot corrosion of Yb₂Zr₂O₇ coated in V₂O₅, was particle-shaped as opposed to rod-like in the Yb₂Zr₂O₇ coated in (Na₂SO₄ + V₂O₅). A dissolutionprecipitation process-based model was put up to discover how hot corrosion products developed. The model could be used to explain the hot corrosion mechanism. It showed that the larger driving force in the hot corrosion of V_2O_5 -coated $Yb_2Zr_2O_7$ contributed to the granular $YbVO_4$ whilst the relatively smaller driving force in the hot corrosion of $(Na_2SO_4 + V_2O_5)$ -coated $Yb_2Zr_2O_7$ resulted in the rod-like $YbVO_4$ [38].

OVERVIEW OF PREVIOUS STUDIES ON HOT CORROSION BEHAVIOUR OF CERAMICS IN MOLTEN SALT

The summary of the overview of earlier studies on the corrosion behaviour of ceramic materials in molten salts is presented in Table 1. Many scholars have worked on several aspects of the area to arrive at different conclusions. Guo et al. [39]

studied the degradation disposition of some ceramic materials $(Sm_{0.5}Sc_{0.5})_2Zr_2O_7$ and $Sm_2Zr_2O_7$ in the presence of molten salt $(V_2O_5 + Na_2SO_4)$. They reported that the ceramic material with the composition $Sm_2Zr_2O_7$ was outperformed by the ceramic with the composition $(Sm_{0.5}Sc_{0.5})_2Zr_2O_7$ in terms of hot corrosion resistance to $(V_2O_5 + Na_2SO_4)$ 50 wt.% salt. In another study [40], the thermal barrier oxides with stabilized ZrO_2 doped with Gd_2O_3 and Yb_2O_3 were investigated to unravel their corrosion behaviour in $V_2O_5 + Na_2SO_4$ molten salt. It was observed that the YSZ ceramic material was more susceptible to corrosion in molten salt than GdYb-YSZ ceramic [40].

Table 1. An Overview of previous studies on hot corrosion behaviour of ceramics in molten salts Tabela 1. Pregled prethodnih studija o ponašanju keramike u toploj koroziji u rastopljenim solima

Ceramic materials	Molten salt composition	Temperature range of study and maximum period of exposure	What to deduce from the study	Ref.
(Sm _{0.5} Sc _{0.5}) ₂ Zr ₂ O ₇ & Sm ₂ Zr ₂ O ₇	50 wt.% V ₂ O ₅ & 50 wt.% Na ₂ SO ₄	700-900 °C/4 h	Compared to Sm ₂ Zr ₂ O ₇ , (Sm _{0.5} Sc _{0.5}) ₂ Zr ₂ O ₇ exhibited greater corrosion resistance to Na ₂ SO ₄ +V ₂ O ₅ salt.	[39]
Thermal barrier oxides with stabilized ZrO ₂ doped with Gd ₂ O ₃ and Yb ₂ O ₃	Gd ₂ O ₃ (1 mol%) + +Yb ₂ O ₃ (1 mol%) doped with Y ₂ O ₃ (3.5 mol%) stabilized ZrO ₂ (GdYb-YSZ)	700-1000 °C / 4 h	Despite being vulnerable to molten salt attack, GdYb-YSZ demonstrated greater hot corrosion resistance than YSZ.	[40]
Zirconia with nanostructures	NaVO₃	700 °C/24–192 h	The YSZ-based thermal barrier coatings' hot corrosion deterioration was significantly accelerated by microcracks and open holes in the coatings.	[41]
Zirconia stabilized – yttria	(Na ₂ SO ₄ + V ₂ O ₅) 50 wt.%	1000 °C / 4 h	The columnar holes that EB-PVD created contributed to the structure's degeneration brought on by the molten salts that seeped from the top layer, speeding up the rate of destruction.	[42]
Magnesia (Stabilized)	Lithium (Li ₂ O & LiCl)	300 – 650 °C/ 360 h	The introduction of lithium gave rise to the initiation of sudden cracks and pores.	[43]
Y-Y ₂ Si ₂ O ₇	V ₂ O ₅ & Na ₂ SO ₄	20 h	The sample of the ceramic (Y-Y ₂ Si ₂ O ₇) bearing alumina additive was reported to give outstanding resistance to corrosion in the presence of molten salt.	[44]
(Gd _{0.9} Sc _{0.5}) ₂ Zr ₂ O ₇	VO ₅	700°C-1000°C/4h	At varying temperatures between 700-1000°C, the ceramic material (Gd _{0.9} Sc _{0.5}) ₂ Zr ₂ O ₇ displayed a potential corrosion resistance coating inhibitor.	[45]
BaLa ₂ Ti ₃ O ₁₀	V ₂ O ₅ ,Na ₂ SO ₄ +V ₂ O ₅	900°C/4h	The ceramic material (BaLa ₂ Ti ₃ O ₁₀) having reacted with V ₂ O ₅ and Na ₂ SO ₄ +V ₂ O ₅ appeared to have suitable corrosion resistance in V ₂ O ₅ compared to Na ₂ SO ₄ +V ₂ O ₅ .	[46]

Yb ₂ O ₃ -Gd ₂ O ₃ -Y ₂ O ₃	CaCl ₂ -CaF ₂ -Cao	1500°C- 1600°C/11h	The ceramic material (Yb ₂ O ₃ -Gd ₂ O ₃ -Y ₂ O ₃) sintered at 1500°C for 6h and experienced the highest degradation compared to 1600°C sintered ceramic material (Yb ₂ O ₃ -Gd ₂ O ₃ -Y ₂ O ₃) for 11h	[47]
SrHfO₃	Na ₂ SO ₄ & V ₂ O ₅	1600°C/10h	A thinner corrosion layer at a higher temperature was observed due to the quick formation of corrosion products ($S_rV_2O_6+m-H_fO_{2(s)}+Na_2O$) which increases as temperature increases.	[48]
LaTi ₂ Al ₉ O ₁₉	V ₂ O ₅	700°C-950°C/10h	AIVO ₄ production was the main reaction seen during a two-hour corrosion session at 700°C but was found to be unstable. It largely dissolved at 800 °C, forming Al ₂ O ₃ . This shows the dynamic character of the corrosion reactions as well as the propensity of some phases to transform or alter under certain circumstances.	[49]
AIN	LiF-LiCl-LiBr-Li	500°C/300h	High levels of metallic Li (>0.1 weight per cent) were reported to hasten the corrosion of AlN.	[50]
Ba1-xSxAx (x=0,0.25,0.5,0.75,1)	Na ₂ SO ₄	900°C-1100°C	At elevated temperatures, there was an enhancement in the diffusion of Na into Ba1-xSxAx which increased the rate of corrosion significantly.	[51]
La ₂ Hf ₂ O ₇	Na ₂ SO ₄ +V ₂ O ₅ & Na ₂ SO ₄	900°C-1200°C/4h	The performance of La ₂ Hf ₂ O ₇ in reaction with Na ₂ SO ₄ +V ₂ O ₅ led to a decrease in the thickness of the corrosion product due to an increase in temperature as contact with V ₂ O ₅ was made.	[52]
Cr ₂ AIC max & CoNiCrAIY	Na ₂ SO ₄ +V ₂ O ₅	950°C/30h	It was reported that Cr ₂ AlC Max has a better hot corrosion resistance than the CoNiCrAlY sample.	[53]
YSZ, Gd ₂ Zr ₂ O ₇ & YSZ/Gd ₂ Zr ₂ O ₇	Na ₂ SO ₄ and V ₂ O ₅	1000°C/25h	Double coatings of YSZ/Gd ₂ Zr ₂ O ₇ were found to resist hot corrosion than those of single coatings	[54]
Zirconia	NaVO ₃ & Na ₂ SO ₄	1100°C/ 6h-24h	The formation of a large polyhedron crystal of YVO ₄ and m-ZrO ₂ as corrosion products was obtained after 6h-24h. Also, the volume fraction of ZrO ₂ in 8YSZ ceramics was found to increase after the corrosion test.	[55]
Ba ₂ REA10 ₅	Na ₂ SO ₄ +V ₂ O ₅	900°C-1000°C/ 4h	The Lewis acid-base rule, thermody- namics, and phase diagrams were used as the foundation for the postu- lated hot corrosion mechanisms.	[56]
Yb ₂ Si ₂ O ₇	NaVO₃	1000°C- 1500°C/2h	YbVO ₄ was obtained on a bulk surface at 1000°C-1200°C owing to the acidity of the corrosion medium and a corrosion product (Yb ₂ Si ₂ O ₅) was obtained.	[57]

Wang et al. [41] investigated how the particle size at the nanoscale can influence the corrosion of zirconia at 700° C in a NaVO₃ corrosive environment. They reported that the samples (n-

YSZ) powders having 30 nm grain sizes were vulnerable to hot corrosion. The reason was attributed to specific surface areas that were larger when compared to those of other specimens. In

another study, the hot corrosion of zirconia stabilized by yttria in ($Na_2SO_4 + V_2O_5$) 50 wt.% molten salt at 1000 °C was undertaken [42]. To prevent edge effects, the corrosion salts were said to be applied to the specimen's surfaces at a concentration of 10 mg/cm², about 1.5 mm away from the edge. It was reported that the interface between CoNiAlY bond (metallic) coating and YSZ ceramic coating existed thermally grown oxide layer including surface cracks.

Cho et al. [43] studied the corrosion of magnesia (stabilized) ceramic in molten salt. The molten salt is composed of lithium (Li₂O & LiCl). They reported that although the degradation of the magnesia ceramic occurred at both cyclic and isothermal conditions, the introduction of lithium increased the rate of corrosion to approximately 5 times compared to the initial degradation rate. On the other hand, the impact of sintering on the degradation of Y-Y2Si2O7 ceramics in molten salt (V₂O₅ & Na₂SO₄) has been carried out [44]. The reaction of Y-Y₂Si₂O₇ ceramic with the following additives: Al, Li, and Mg resulted in 2 mol.% of Al₂O₃, Li₂O, and MgO, producing three different samples. The ceramic (Y-Y₂Si₂O₇) sample with the alumina addition was said to have exceptional corrosion resistance in the presence of molten salt [44]. Zhang et al. [45] identified the effect of molten salt (V₂O₅) on ceramic material, bearing the composition, $Gd_{0.9}Sc_{0.5})_2Zr_2O_7$. They discovered that the corrosion product (Sc₂O₃), obtained following the interaction between the ceramic material (Gd_{0.9}Sc_{0.5})₂Zr₂O₇ and the molten salt (V₂O₅) possessed an outstanding resistance to corrosion due to the molten salt.

Liu et al. [46] discovered the influence of molten salts (Na₂SO₄, and V₂O₅) on the ceramic material (BaLa₂Ti₃O₁₀). According to the results of this investigation, BaLa₂Ti₃O₁₀ ceramic, a potential candidate for Thermal Barrier Coating, exhibits greater resistance to V2O5 salt corrosion despite having strong resistance to both salts. On the other hand, the effect of molten salt (CaCl₂-CaF₂-CaO) on ceramic material (Yb2O3-Gd2O3-Y2O3) was carried out by Kim et al. [47]. They reported that the exposure time could act as a driving force for the penetration of the molten salt (CaCl2-CaF2-CaO) which further shows a higher degradation rate of the sample (Yb₂O₃-Gd₂O₃-Y₂O₃) sintered at 120h compared to the sample sintered at approximately 6h-11h.

Gu et al. [48] studied the influence of molten salts (Na_2SO_4 , V_2O_5 and the combination of Na_2SO_4 + V_2O_5 (in the molar proportion of 1:1) on the microstructure of the ceramic material (SrHfO₃). A schematic illustration of what subsequently occurred and the nature of the corrosion product

can be found in Fig. 1. The authors reported that while the thickness of the corrosion scale increased in mixed molten salt, the thickness of the corrosion layer reduced as the temperature increased in contact with V_2O_5 [48]. A thinner corrosion layer at a higher temperature was observed due to the quick formation of corrosion products ($S_rV_2O_6+m-H_fO_{2(s)}+Na_2O$) which increases as temperature increases.

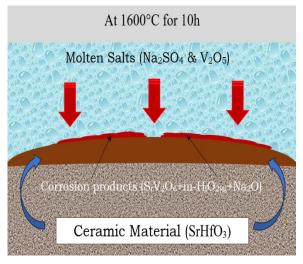


Figure 1. Formation of corrosion products $(S_rV_2O_6+m-H_rO_{2(s)}+Na_2O)$ by the exposure of $SrHfO_3$ in Na_2SO_4 & V_2O_5

Slika 1. Formiranje produkata korozije (SrV₂O₆+m-HfO₂(s)+Na₂O) izlaganjem SrHfO3 u Na₂SO₄ i V₂O₅

Evaluation of molten salt (V2O5) on ceramic material (LaTi₂Al₉O₁₉) has been carried out [49]. It was reported that the temperature and length of the corrosion process were found to dictate the nature of the corrosion products and microstructures of the LaTi₂Al₉O₁₉ ceramic. Variable reactions and phase formations were produced by altering temperatures and periods. In another interesting study, the AIN ceramics' corrosion behaviour in 500 [50]. °C LiF-LiCI-LiBr-Li molten salt researchers observed that high levels of metallic Li (>0.1 weight per cent) increased the corrosion of AIN. The interaction between Li and the YAG (yttrium aluminium garnet) found in AIN was reported to cause the observed accelerated corrosion. This implies that the existence of high Li concentrations encourages a corrosive environwhich damages AIN [50]. In a molten Na₂SO₄ environment, Jiang et al. [51] investigated the hot corrosion behaviour of barium-strontium aluminosilicates. They reported that the diffusion reaction of the Ba/Sr and Na cations had an impact on the hot corrosion behaviour of $B_{1-x}\dot{S_x}AS$ compounds in a molten Na₂SO₄ environment. This mechanism greatly aided the degradation process. Additionally, the order of cation migration during hot corrosion was impacted by the difference in atomic radius between strontium (Sr) and barium (Ba). It was discovered that strontium cations diffused out of the B_{1-x}S_xAS structure before barium cations, suggesting that strontium should exit the structure first. A study to determine performance of La₂Hf₂O₇ ceramic when exposed to the molten salt (sulphate-vanadate) at 900°C-1200°C has been carried out [52]. They discovered nanoparticles (non-needle-like) were present and the corrosion rate was directly proportional to the increment in the temperature accompanied by LaVO₄ crystalline grain growing noticeably. Furthermore, the thickness of the corroded layer was reported to increase with an increase in corrosion temperature. A study on the behaviour of the Cr₂AIC max compound in the presence of V₂O₅ & Na₂SO₄ molten salts has been carried out [53]. The researchers discovered that under the test circumstances, the Cr₂AlC MAX-phase samples demonstrated greater hot corrosion resistance than the CoNiCrAIY samples.

Ozgurluk et al. [54] experimented with the reaction of $YSZ/Gd_2Zr_2O_7$ when exposed to vanadate salt (V_2O_5) at $1000^{\circ}C$. They reported that after the conclusion of the experiment (25h) within 5h cycles at $1000^{\circ}C$, significant deterioration and spillage on the YSZ coatings was observed. On the other hand, a study on the effect of the presence of $NaVO_3+Na_2SO_4$ on Zirconia ceramics at $1100^{\circ}C$ has been conducted [55].

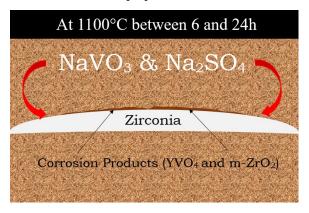


Figure 2. Effect of the presence of NaVO₃+Na₂SO₄ on Zirconia ceramics at 1100 °C

Slika 2. Uticaj prisustva NaVO₃+Na₂SO₄ na cirkonijum keramiku na 1100 °C

It was observed that Na_2SO_4 was a promoting factor of chemical reactions which increased the rate of corrosion. The process yielded Yttrium orthovanadate (YVO₄) and monoclinic zirconia dioxide (m-ZrO₂) as the corrosion products. Another study to evaluate the hot corrosion mechanisms of Ba_2REA10_5 when exposed to $Na_2SO_4+V_2O_5$ was successfully carried out [56].

According to the study, Ba₂REAIO₅ formed a reaction layer with a thickness of around 80 µm after being exposed to a temperature of 1000 °C for 4 hours. This implies that at this higher temperature, the material is more prone to corrosion. Finally, Zhang et al. [57] undertook a study to determine the behaviour of Yb2Si2O7 ceramics when exposed to molten salt (NaVO₃) at 1000°C-1500°C. They observed that the reaction between Yb2Si2O7 and NaVO3 produced rodshaped YbVO₄ (ytterbium vanadate) grains at temperatures between 1000 and 1200 °C. As the temperature increased, YbVO₄ concentration fell until it vanished at 1300 °C. The hot corrosion product, Yb₂SiO₅ (ytterbium silicate), showed a polyhedral form at temperatures of 1300 °C or higher. With an increase in temperature, the corrosion products contained more Yb₂SiO₅.

4. CONCLUSION

From the present study on the overview of how ceramic materials behave when exposed to molten salt environments, the following conclusions may be drawn:

- The composition of ceramics has a big impact on corrosion resistance. In the presence of molten salt (V₂O₅ + Na₂SO₄), studies have shown that the ceramic material (Sm_{0.5}Sc_{0.5})₂Zr₂O₇ performed better than Sm₂Zr₂O₇ in terms of hot corrosion resistance. This emphasizes how crucial it is to choose ceramic compositions carefully to increase corrosion resistance in particular situations.
- The size of the particles affects corrosion behaviour. In a NaVO₃ corrosive environment, it was reported that zirconia ceramics with smaller particle sizes (30 nm grain sizes) were more susceptible to heat corrosion. This shows that certain surface areas, affected by particle size, contribute to the susceptibility of ceramic materials to corrosion.
- Additives can improve corrosion resistance. Studies have shown that using alumina (Al₂O₃) as an additive to Y-Y₂Si₂O₇ ceramics produced remarkable resistance to corrosion in molten salt (V₂O₅ & Na₂SO₄). This suggests that adding carefully chosen additives might enhance the corrosion resistance of ceramic materials.
- The alumina layer has been found to exhibit resistance to corrosion. The identification of an alumina layer (corrosion product) with exceptional corrosion resistance in a variety of molten salt conditions raises the possibility that this layer may play a significant role in determining how quickly ceramic materials

corrode. An alumina layer can serve as a barrier of defence against further deterioration.

These findings show the intricate interactions between the composition of ceramic materials, particle size, additives, molten salt composition, and the production of protective corrosion products that affect how they react to corrosion and how resistant they are to it. The development of corrosion-resistant ceramics for specific applications requires an understanding of these aspects.

5. KNOWLEDGE GAPS AND POTENTIAL RESEARCH AREAS

Following the overview of the behaviour of ceramics during hot corrosion in a variety of molten salts, the following areas have been identified as knowledge gaps. Additionally, some recommendations for possible future study areas have been made:

- Although the reviewed papers investigated the hot corrosion behaviour of several ceramics in molten salts, they did not go into detail about the precise corrosion mechanisms at play. It would be beneficial to comprehend the exact mechanisms by which corrosion occurs in various ceramic materials to help produce corrosion-resistant coatings or materials.
- The reviewed papers included the maximum exposure time for each investigation; however, they gave no details on how long the ceramic materials will last in hot corrosion environments. To determine if ceramics are suitable for use in practical applications, it would be crucial to look at the stability and degradation of ceramics, especially in the presence of sulphate-based molten salts over long periods.
- The reviewed studies discussed several ceramic compositions and how resistant they were to heat corrosion. The tuning of composition characteristics, such as dopants, additives, or material processing strategies to improve corrosion resistance, was not covered, nevertheless. Additional investigation is required to determine the composition's effect on hot corrosion behaviour and the best compositions for various applications.
- The major emphasis of many of the reviewed studies was on the temperature and the composition of molten salts as the main determinants of hot corrosion behaviour. The corrosion process, however, can also be impacted by other environmental variables such as gas atmospheres, contaminants, and thermal cycling. A deeper knowledge of the hot corrosion behaviour of ceramics would result

- from examining the combined influence of these elements.
- The reviewed studies largely involved qualitative analyses of hot corrosion behaviour, concentrating on the relative resistance or susceptibility of various ceramics. To evaluate the performance of ceramics under hot corrosion circumstances, quantitative measurements of corrosion rates, mass loss, or other degradation characteristics would offer more accurate and comparable information.

It would be easier to create better corrosionresistant materials if these information gaps were filled, which would lead to a better understanding of hot corrosion behaviour in ceramics.

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IZVOD

PREGLED KOROZIONOG PONAŠANJA KERAMIČKIH MATERIJALA U OKRUŽENJIMA RASTOPLJENE SOLI

Ova studija je pregledala prethodne studije između 2015. i 2021. o tome kako su keramički materijali degradirali u prisustvu rastopljene soli. Takođe, ispitani su procesi otpornosti na koroziju različitih keramičkih kompozicija izloženih različitim sastavima rastopljene soli i temperaturama. Rezultati nude važne nove uvide u varijable koje utiču na koroziono ponašanje keramike i proizvodnju proizvoda korozije. Prijavljeni rezultat otkriva da je keramički materijal sastava (Sm0.5Sc0.5) $_2$ Zr2O $_7$ imao bolji učinak od Sm $_2$ Zr $_2$ O $_7$ u pogledu otpornosti na toplu koroziju u rastopljenoj soli (V $_2$ O $_5$ + Na $_2$ SO $_4$). Takođe, prijavljeno je da na ponašanje korozije utiče veličina čestica. Značajno je da je cirkonijum (n-ISZ) sa veličinom zrna na nanoskali bio podložniji vrućoj koroziji, što se objašnjava povećanim specifičnim površinama. S druge strane, utvrđeno je da sinterovanje i aditivi povećavaju otpornost na koroziju. Otpornost I-I $_2$ Si $_2$ O $_7$ keramike na koroziju u (V $_2$ O $_5$ + Na $_2$ SO $_4$) rastopljenoj soli je poboljšana dodatkom glinice. Rezultati ovih istraživanja pomažu nam da razumemo kako korozija funkcioniše i šta utiče na podložnost keramičkih materijala propadanju u rastopljenim slanim medijima. Ove informacije mogu da usmere stvaranje keramičkih materijala otpornijih na koroziju za upotrebu u okruženjima sa visokim temperaturama ili energetskim sistemima na bazi rastopljene soli, između ostalih korozivnih upotreba.

Ključne reči: vruća korozija, glinica, izmenjivači toplote, nitridi, cirkonijum

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